



## Promotional removal of HCHO from simulated flue gas over Mn-Fe oxides modified activated coke



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### ABSTRACT

A series of  $Mn_xFe_y/AC$  catalysts synthesized by impregnation method were investigated on the efficient and stable removal of HCHO in the fix-bed reactor. Extensive characterizations, BET, SEM, XRD,  $H_2$ -TPR, XPS and FT-IR, were conducted to study the physicochemical properties, HCHO oxidation and surface reaction of catalysts. The optimal  $Mn_{0.75}Fe_{6.02}/AC$  showed enhanced HCHO removal efficiency of 98.30%, as well as excellent performance for simultaneous removal of HCHO (89.96%) and  $Hg^0$  (77.51%).  $NO$  and  $SO_2$  balanced in  $N_2$  would inhibit the removal of HCHO, while the addition of 6%  $O_2$  weakened the negative effect of  $SO_2$  and  $NO + 6\% O_2$  facilitated the removal of HCHO. Besides, the slight promotion effect of water vapor was contributed to the regeneration of consumed  $-OH$  via the activation of surface oxygen by adsorbed  $H_2O$ . Characterization results indicated that  $Mn_{0.75}Fe_{6.02}/AC$  possessed larger BET surface area, well-developed porosity and better dispersion of active components.  $Mn_{0.75}Fe_{6.02}/AC$  exhibited higher reducibility due to the synergistic effect between  $MnO_x$  and  $FeO_x$ , and the interaction between Mn-Fe oxides and AC support. At the same time, the oxygen-containing functional groups (C-O, COO), abundant active surface oxygen and  $-OH$  facilitated both adsorption and oxidation of HCHO. Besides, the formate and carbonate intermediates formed on the surface of  $Mn_{0.75}Fe_{6.02}/AC$  in HCHO removal process, which could be further oxidized into  $CO_2$  and  $H_2O$ . On the basis of above investigations, the mechanism of enhanced HCHO catalytic removal over  $Mn_xFe_y/AC$  was proposed.

### 1. Introduction

Volatile organic compounds (VOCs) released from anthropogenic sources are an important contributor of atmospheric pollution. Coal combustion has been considered as a major source of anthropogenic VOCs emissions due to the enormous demand of coal in industrial manufacturing. As one of the hazardous VOCs, formaldehyde (HCHO) has been widely considered as a great threat to the human health [1,2]. Besides, HCHO can also react with  $NO_x$  and make contribution to photochemical smog due to its strong photochemical activity [3,4]. Therefore, it is urgent to develop efficient technologies for HCHO removal.

To meet the stringent environmental regulations, various technologies, such as adsorption [5], condensation [6], photo-catalysis [7], catalytic oxidation [8], plasma technology [9], and biological filtration [10], have been developed to eliminate the emission of HCHO. Among them, catalytic oxidation is considered as a promising technology due

to its saving of energy and the minimization of the generation of toxic by-products [11].

In recent decades, the main catalysts for adsorption and catalytic oxidation of HCHO are supported noble metals or transition metal oxides. These catalysts are highlighted to be an effective and promising technology with high efficiency and low energy consumption. In particular, various metal oxides supports, including  $TiO_2$  [12],  $MnO_x$  [13],  $CeO_2$  [14],  $FeO_x$  [3,15],  $ZrO_2$  [16],  $Al_2O_3$  [17] and the metal composites like  $CeO_2-Co_3O_4$  [18],  $MnO_x-CeO_2$  [19] and  $In_2O_3-SnO_2$  [20] are usually used for HCHO oxidation, and could facilitate the activity of deposited active components due to their high dispersion, strong redox properties of the support and the synergy effect with supported metal oxides. However, several shortcomings of current employed metal oxides supports, such as non-uniform structure, uncontrollable size and shape, as well as high cost, have prevented their application. Therefore, porous supports with regular structure have attracted much attention [21–26]. As a kind of carbon based material with porous structure,

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activated coke (AC) possesses considerable environmental and economic benefits on account of its rich source, high mechanical strength, lower cost and better regeneration performance [27,28]. Compared with activated carbon, AC is comprised mainly of macropore and mesopore instead of micropore, which reduces significantly the impact of internal diffusion on the general rate of adsorption and catalytic processes. Hence, AC is usually considered as a suitable support used for the decomposition of VOCs into  $H_2O$  due to its hydrophilic character. Additionally, AC could be used for the removal of numerous pollutants such as  $SO_2$  [29],  $H_2S$  [30],  $NO_x$  [31] and  $Hg^0$  [32–34]. The literatures [35,36] reported that AC has been also exploited in treating organic pollutants in wastewater from industrial process.

Nevertheless, the removal capacity of virgin AC towards HCHO is restricted by its limited physicochemical property such as activated site and surface area. It has been reported that the carbon-based material loaded with metal oxides possessed both high adsorption and catalytic ability [37]. Therefore, the modification of AC by metal oxides seems to be a potential approach to improve the activity of AC to a great extent [32,38,39].

As previously reported [8,40,41], some non-noble metal oxides with fast electron transference and variable valences, such as manganese oxides and iron oxides, have been used as active phase owe to their low cost, wide availability and high resistance to sulfur and chlorine. Significantly, manganese-based catalysts have been focused as promising, inexpensive, environmental friendly and effective catalysts for HCHO removal. For example, large oxygen storage capacity of manganese oxides caused by the ability to switch oxidation states ensures higher activity for VOCs removal [42]. As is reported, manganese oxides have been proved to be more effective than some transition metal oxides, such as  $TiO_2$ ,  $CeO_2$ , or even some binary oxides for HCHO removal [43]. It is well known that Mn has variable valence states from  $-3$  to  $+7$  due to its electronic structure  $3d^54s^2$  [8]. On one hand, the manganese oxides are able to mobilize electrons, thus generating the mobile-electron environment required by redox reaction. On the other hand, their redox properties are greatly enhanced with incorporation of other elements [44]. Currently, Fe was widely applied and exhibited good performance for VOCs oxidation. It was demonstrated that  $Fe^{3+}$  sites could facilitate the formation of  $O_2^{2-}$  species that are vital for the activation and oxidation of VOCs [45]. The capability of Fe containing oxide to generating active oxygen species has also been highlighted [46]. Therefore, ideas for the co-modification of  $MnO_x$  and  $FeO_x$  over AC ( $Mn_xFe_y/AC$ ) to improve the activity are encouraged. Unfortunately, there were still few reports about using  $Mn_xFe_y/AC$  for the removal of HCHO from simulated flue gas. In addition, the mechanism of enhanced HCHO removal over the catalyst was still not very clear.

Therefore, in this paper, a series of  $Mn_xFe_y/AC$  samples were investigated for HCHO removal, with emphasis on four aspects: (i) removal performance of HCHO over  $Mn_xFe_y/AC$  and their potential in practice application; (ii) the simultaneous removal of  $Hg^0$  and HCHO; (iii) the physicochemical and structural properties; (iv) the mechanism of HCHO removal. The objective of the work is to give an in-depth exploration in the relationship between the physicochemical properties and catalytic activity of  $Mn_xFe_y/AC$ , and develop an eco-friendly and effective catalyst for HCHO removal.

## 2. Experimental section

### 2.1. Samples preparation

A series of  $Mn_xFe_y/AC$  (where  $x$  denotes the  $Mn/(Mn + Fe)$  molar ratio in Mn-Fe oxides, while  $y$  denotes the total mass percent of metal Mn-Fe on  $Mn_xFe_y/AC$ ) samples have been prepared by equivalent volume impregnation. Initially, the virgin AC ( $D = 3$  mm and  $L = 4\text{--}6$  mm, Alxa League Ke'xing Carbon industry) was washed with deionized water repeatedly and then dried at  $105^\circ C$  for 12 h. Secondly, appropriate amounts of 50 wt.%  $Mn(NO_3)_2$  solution and  $Fe(NO_3)_3 \cdot 9H_2O$

were dissolved in deionized water and the AC was impregnated in the solution at room temperature for 24 h. For example, as for  $Mn_{0.75}Fe_{3.30}/AC$ , 2.156 mL 50 wt.%  $Mn(NO_3)_2$  solution and 1.2471 g  $Fe(NO_3)_3 \cdot 9H_2O$  were dissolved into deionized water as impregnation solution. The obtained precursor was then dried at  $105^\circ C$  for 12 h and finally calcined at  $500^\circ C$  for 4 h under  $N_2$  atmosphere. The metal loading value, described as  $y$ , was equal to 0.65%, 1.97%, 3.30%, 4.65%, 6.02% and 7.42%. Besides, to identify the effect of the molar ratio of Mn/ (Mn + Fe) on physicochemical properties of samples, the value of  $x$  was set as 0, 0.15, 0.3, 0.5, 0.75, 0.9, 1.

### 2.2. Samples characterization

To study the basic characters of virgin AC, the proximate analysis of virgin AC were measured by Chinese National standards (GB/T 212-2008). And the ultimate analysis of virgin AC was conducted using the Elementar Analysensystem GmbH vario (Elementar Ltd Corp, Germany).

The actual content of metal doping over AC was measured by Inductively Coupled Plasma-Atomic Emission Spectrometry (ICP-AES, SPECTRO BLUE SOP, Germany).

The pore structure including the Brunauer-Emmett-Teller specific surface area ( $S_{BET}$ ), pore volume and average pore size was determined by standard nitrogen adsorption on a Micromeritics Tri-star 3020 analyzer (Micromeritics Instrument Crop, USA). And samples were previously degassed at  $120^\circ C$  before BET measurement.

The scanning electron microscopy (SEM) micrographs ( $\times 10,000$ ) of all samples were collected on a JSM-6700F (Japan) instrument.

The X-ray diffraction (XRD) patterns were obtained on a Rigaku rotaflex D/Max2500 powder diffractometer (Rigaku, Japan) at room temperature with  $Cu-K\alpha$  radiation at  $40$  KV and  $250$  mA. The diffraction line positions were determined with a step of  $0.02^\circ$  in a scanning range of  $10\text{--}80^\circ$  ( $2\theta$ ).

To identify the redox properties of samples, the temperature-programmed reduction of hydrogen ( $H_2$ -TPR) was performed on an AutoChem 2090 automated chemisorption analyzer (Micromeritics Instrument Crop, USA). For the analysis, about  $100$  mg of samples were pretreated under  $N_2$  flow at  $300^\circ C$  for 30 min and then cooled down to the room temperature. The reduction measurement was carried out from the room temperature to  $900^\circ C$  with a heating rate of  $10^\circ C/min$  under  $5\%$   $H_2$ -Ar ( $40$  mL/min) atmosphere.

The X-ray photoelectron spectroscopy (XPS) analyses were conducted on a K-Alpha 1063 spectrometer (Thermo Fisher Scientific, UK) to study the chemical states of the constituent elements. All spectra were recorded using an  $Al\ K\alpha$  X-ray radiation source. The charge-shifted spectra were referenced to C 1s line position at  $284.6$  eV.

The Fourier Transform Infrared Spectroscopy (FT-IR) was collected in range of  $500\text{--}4000\text{ cm}^{-1}$  on a FTIR-8400 S IRprestige-21 (SHIMADZU, Japan) apparatus at a spectral resolution of  $2\text{ cm}^{-1}$  to characterize the intermediates and products during HCHO removal on sample surface. Prior to each measurement, the sample was pretreated at  $250^\circ C$  in a pure  $N_2$  gas flow of  $500$  mL/min for 1 h for the removal of impurity and moisture adsorbed on sample and subsequently cooled down to room temperature. Then the pretreated sample was exposed to the corresponding gas atmosphere of HCHO (120 ppm HCHO balanced in  $N_2$ ) at room temperature and  $HCHO + O_2$  (120 ppm HCHO, 6%  $O_2$  and balanced  $N_2$ ) at room temperature and at optimal reaction temperature, for 2 h to be saturated, respectively.

### 2.3. Experimental setup and procedure

The experiments of the removal of HCHO on  $Mn_xFe_y/AC$  were carried out in a continuous flow-type apparatus, consisting of a continuous simulated flue gas supplying system, a fixed-bed quartz reactor (52 mm inner diameter  $\times$  870 mm length) and a gas analyzer system, as shown in Fig. 1. The feed of gas-phase HCHO to the reactor was

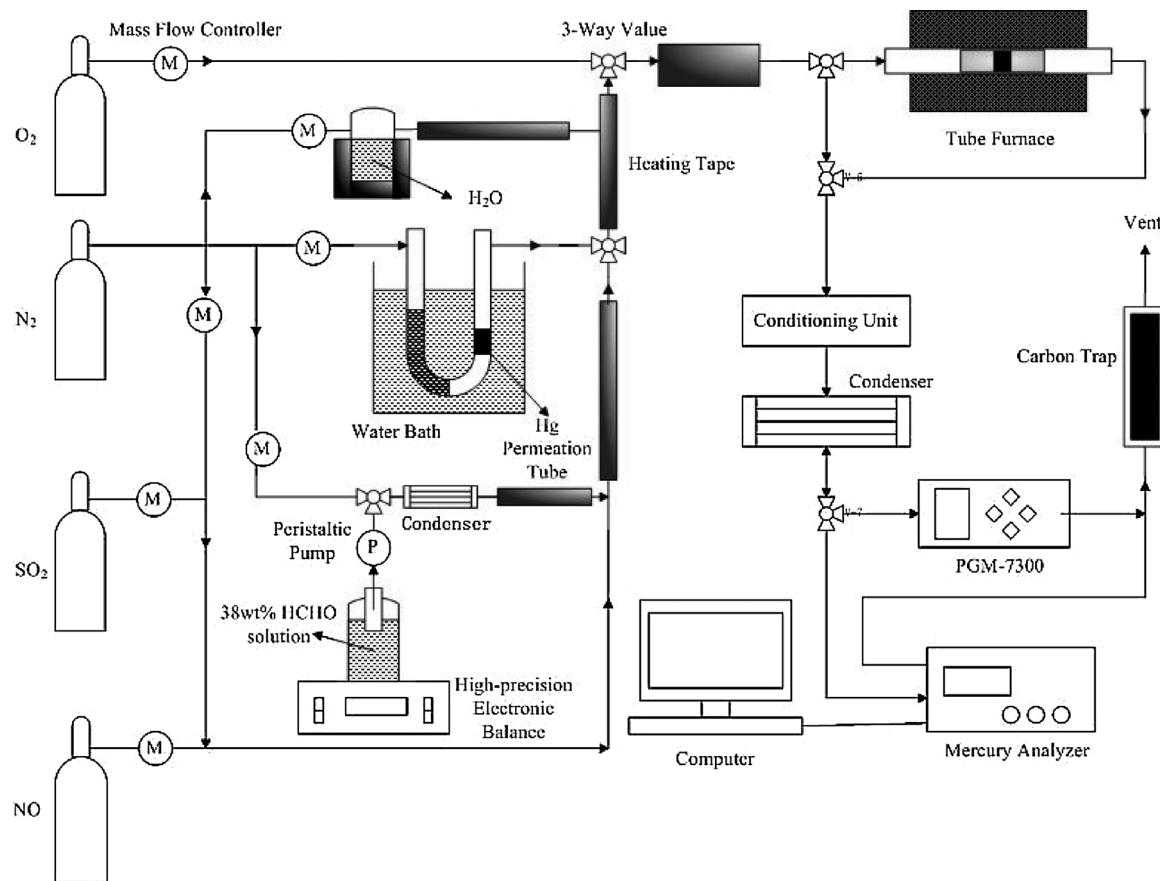


Fig. 1. Schematic diagram of the experimental setup.

performed by controlling a peristaltic pump to inject 38 wt.% formaldehyde solution into Teflon tube wrapped with temperature-controlled heating tape. Then 100 mL/min N<sub>2</sub> took generated gas-phase HCHO alone through the condensing apparatus to remove the water vapor, and then mixed with the simulated flue gas. The Hg<sup>0</sup> vapor-generating device was made up of an elemental mercury permeation tube (VICI Metronics, USA) and a water bath with a tunable temperature. A feed of 120 ppm HCHO, 6 vol.% O<sub>2</sub>, 90 µg/m<sup>3</sup> Hg<sup>0</sup> (when used) and the balance of N<sub>2</sub> were controlled by mass flow controllers respectively and the total flow rate of the flue gas was maintained at 500 mL/min, corresponding to a gas hourly space velocity (GHSV) of 2500 h<sup>-1</sup>. For each test, 20 g sample was placed in the middle of the quartz tube horizontally. The concentration of inlet (HCHO<sub>in</sub>) and outlet (HCHO<sub>out</sub>) HCHO were measured by PGM7300 analyzer (RAE, USA), and the selectivity of HCHO over Mn<sub>x</sub>Fe<sub>y</sub>/AC was calculated combining with the concentration of outlet CO<sub>2</sub> which was monitored by PGA-650 (Phymetrix, USA). The inlet ([Hg<sup>0</sup>]<sub>in</sub>) and outlet Hg<sup>0</sup> ([Hg<sup>0</sup>]<sub>out</sub>) concentration were measured using an online RA-915M mercury analyzer (LUMEX Ltd, Russia). The oxidation of mercury was measured by the difference of Hg<sup>0</sup> concentration through the mercury transformation system and [Hg<sup>0</sup>]<sub>out</sub>, as depicted in Tao et al. [32]. After the desired inlet concentration has been established for 30 min, the experimental gas stream was introduced to pass through each sample until the HCHO<sub>out</sub> reached stable in pre-experiment. Prior to testing the HCHO and Hg<sup>0</sup> removal efficiency, blank test was conducted to avoid any influence of the reactor. And the reaction time for each test was 6 h.

The removal efficiency of HCHO (E<sub>HCHO</sub>) and conversion to CO<sub>2</sub> (S<sub>C</sub>) was calculated from the equation below:

$$E_{HCHO}(\%) = \frac{\Delta HCHO}{HCHO_{in}} = \frac{HCHO_{in} - HCHO_{out}}{HCHO_{in}} \times 100\% \quad (1)$$

$$S_C(\%) = \frac{CO_2}{HCHO_{in} \times E_{HCHO}} \times 100\% \quad (2)$$

where HCHO<sub>in</sub> and HCHO<sub>out</sub> represent HCHO concentration (ppm) at the inlet and outlet of the fixed-bed reactor, respectively.

The mercury removal efficiency (E<sub>Hg</sub>) was defined as:

$$E_{Hg}(\%) = \frac{\Delta Hg^0}{[Hg^0]_{in}} = \frac{[Hg^0]_{in} - [Hg^0]_{out}}{[Hg^0]_{in}} \times 100\% \quad (3)$$

$$E_{oxi}(\%) = \frac{[Hg^0]_{oxi}}{[Hg^0]_{in}} \times 100\% \quad (4)$$

where [Hg<sup>0</sup>]<sub>in</sub> is the inlet Hg<sup>0</sup> concentration and [Hg<sup>0</sup>]<sub>out</sub> is the outlet Hg<sup>0</sup> concentration (µg/m<sup>3</sup>), and [Hg<sup>0</sup>]<sub>oxi</sub> is the outlet Hg<sup>2+</sup> concentration (µg/m<sup>3</sup>).

### 3. Results and discussion

#### 3.1. Textural, structural and morphologic characterizations

##### 3.1.1. Proximate, ultimate analysis and ICP-AES

As shown in Table 1, the basis characters of virgin AC were investigated by proximate and ultimate analysis. The ultimate composition of virgin AC is 86.88% C + 0.70% H + 11.11% O + 0.68% N + 0.63% S, and the proximate analysis indicated that the virgin AC was comprised of 2.57% moisture, 12.67% ash, 22.74% volatile and 62.02% fixed carbon.

The actual content of metal in different samples was measured using ICP-AES. As summarized in Table 2, the actual content of metal was slightly lower than the theoretical value for all samples. The slight difference might be attributed to the inevitable loss of metal nitrate precursor during the impregnation process.

**Table 1**

Proximate and ultimate analysis of virgin AC (air-dry basis).

Samples	Proximate analysis (wt.% ad)				Ultimate analysis (wt.% ad)				
	M	A	V	FC <sup>a</sup>	C	H	O <sup>a</sup>	N	S
Virgin AC	2.57	12.67	22.74	62.02	86.88	0.70	11.11	0.68	0.63

<sup>a</sup> by difference M % + A % + V % + FC<sup>a</sup> % = 100%; C % + H % + O<sup>a</sup> % + N% + S % = 100%.

**Table 2**

The content (wt.%) of metal in samples obtained by ICP-AES.

Sample	Mn content (wt.%)		Fe content (wt.%)	
	nominal content	actual content	nominal content	actual content
Fe <sub>3.30</sub> /AC	—	—	3.30	3.18
Mn <sub>0.30</sub> Fe <sub>3.30</sub> /AC	0.98	0.87	2.32	2.21
Mn <sub>0.50</sub> Fe <sub>3.30</sub> /AC	1.64	1.51	1.66	1.58
Mn <sub>0.75</sub> Fe <sub>3.30</sub> /AC	2.46	2.32	0.84	0.79
Mn <sub>3.30</sub> /AC	3.30	3.21	—	—
Mn <sub>0.75</sub> Fe <sub>0.65</sub> /AC	0.49	0.41	0.17	0.10
Mn <sub>0.75</sub> Fe <sub>6.02</sub> /AC	4.49	4.34	1.53	1.41

### 3.1.2. BET

Textural properties of virgin AC and Mn<sub>0.75</sub>Fe<sub>y</sub>/AC, including BET specific surface area, pore volume and pore size, were investigated by N<sub>2</sub> adsorption-desorption measurement. As shown in Table 3, the co-modification of MnO<sub>x</sub> and FeO<sub>x</sub> could increase the surface area and pore volume of AC to a certain extent. It could be found that the Mn-Fe modified AC except Mn<sub>0.75</sub>Fe<sub>7.42</sub>/AC, possessed higher BET surface area and pore volume than virgin AC. At the same time, BET surface area and pore volume increased as the loading of metal Mn-Fe increased from 0.65% to 6.02%, which increased from 237.62 m<sup>2</sup>/g and 0.121 cm<sup>3</sup>/g to 272.21 m<sup>2</sup>/g and 0.145 cm<sup>3</sup>/g, respectively. It might be attributed to a number of new pores formed during impregnation and calcination [34]. Nevertheless, with further increasing the active components loadings, a marked reduction in the BET surface area and pore volume was observed. According to Tao et al. [32], the decrease might be ascribed to that an excess of metal oxides loaded on AC formed the agglomeration to cover the external surface of AC, thus blocking a number of pores. In particular, Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC possessed the highest specific surface area and pore volume. In general, larger BET surface area the material possessed, there were more active sites and larger interface for the adsorption and the following oxidation HCHO on Mn<sub>x</sub>Fe<sub>y</sub>/AC catalysts, which gave rise to a high catalytic activity [47].

### 3.1.3. SEM

The external morphology and microstructure of selected samples were characterized by SEM, as depicted in Fig. 2. Compared with virgin AC, the morphological characteristics of AC have been changed after loading with Mn-Fe oxides, which were homogeneously dispersed on the surface of AC at micrometric scale. In addition, more and more particles were observed over surface of samples with the increase of Mn-Fe loading below 3.30%. It is noted that rough surface of

Mn<sub>0.75</sub>Fe<sub>4.65</sub>/AC and Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC exhibited some interlamellar structure (Fig. 2(e)–(f)), especially for Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC, thus increasing its BET surface area, which endowed the samples more efficient in adsorbing HCHO and other gaseous components [48]. However, compared with Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC, there were more agglomerates rather than distinct interlamellar structure observed over Mn<sub>0.75</sub>Fe<sub>7.42</sub>/AC, which might be the result of excess active components deposition. The appearance of agglomerates led to the blockage of pores and a lower specific surface area, which was in accordance with BET results (Table 3).

### 3.1.4. XRD

As shown in Fig. 3A, the XRD measurement was performed to identify the crystal structures of the Mn<sub>0.75</sub>Fe<sub>y</sub>/AC samples with different metal loadings, together with virgin AC. It could be seen that the diffraction lines for SiO<sub>2</sub> (2θ = 28.899°, 32.220° and 36.040°, PDF-ICDD 18-1170) were detected over virgin AC. On one hand, the primary peaks at 2θ = 26.603°, 44.464° (PDF-ICDD 25-0248) corresponding to carbon were observed over all samples, revealing that the microstructure of AC was remained to a certain extent after impregnation. On the other hand, with the doping of Mn-Fe oxides over AC, the intensity of the characteristic peaks of carbon decreased with increasing metal oxides content, suggesting that the loading of metal oxides over AC changed the natural structure of AC. The results above also could be inferred that the interaction may exist between Mn-Fe oxides and AC, as demonstrated in BET, SEM and XPS. Interestingly, the diffraction peaks at 2θ = 32.315°, 36.085°, 44.440° and 59.840° were ascribed to Mn<sub>3</sub>O<sub>4</sub> (PDF-ICDD 24-0734), and the diffraction peak at 2θ = 40.547° was attributed to MnO (PDF-ICDD 07-0230), while no related peaks of Fe<sub>2</sub>O<sub>3</sub> or FeO were observed on the surface of Mn<sub>0.75</sub>Fe<sub>y</sub>/AC, suggesting that the content of iron oxides was too low to form crystalline phase according to monolayer dispersion theory [49]. Besides, the particle size of metal oxides increased with the increasing loading value. The particle size of Mn<sub>3</sub>O<sub>4</sub> estimated by the Scherrer equation and MDI Jade was 12.32 nm and 12.09 nm for Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC and Mn<sub>0.75</sub>Fe<sub>4.65</sub>/AC, respectively. Compared with Mn<sub>7.42</sub>/AC, after the doping of FeO<sub>x</sub>, the intensity of peaks assigned to Mn<sub>3</sub>O<sub>4</sub> weakened distinctly (Fig. 3B), demonstrating the existence of FeO<sub>x</sub> led to the lower crystallinity or smaller crystallite size of MnO<sub>x</sub>, and presumably the subsequent high catalytic activity.

### 3.2. Redox properties

#### 3.2.1. H<sub>2</sub>-TPR

As is recognized, the catalytic oxidation of VOCs on catalysts where metal oxides act as the active components occurred through the oxidation of organic molecule by the lattice oxygen originating from metal oxides and subsequently the reoxidation of the catalyst by gas-phase oxygen [50]. To keep the mechanism proceeding, the redox pairs of supported metal oxides in different oxidation states are essential during the elimination of VOCs. As illustrated in Fig. 4, for virgin AC, the H<sub>2</sub>-TPR profile revealed two overlapping peaks characterized by apparent maxima at 485 and 658 °C approximatively. The first peak at about 485 °C might be assigned to the reduction of surface oxygen species, and the higher peak corresponds to the gasification of the carbon support [51,52]. However, after the loading of MnO<sub>x</sub>, Mn<sub>6.02</sub>/AC

**Table 3**

BET surface and pore parameters of the different samples.

Samples	BET surface area (m <sup>2</sup> /g)	Pore volume (cm <sup>3</sup> /g)	Average pore diameter (nm)
AC	237.62	0.121	2.038
Mn <sub>0.75</sub> Fe <sub>0.65</sub> /AC	248.14	0.125	2.017
Mn <sub>0.75</sub> Fe <sub>3.30</sub> /AC	265.83	0.130	1.948
Mn <sub>0.75</sub> Fe <sub>6.02</sub> /AC	272.21	0.145	2.127
Mn <sub>0.75</sub> Fe <sub>7.42</sub> /AC	218.29	0.117	2.155

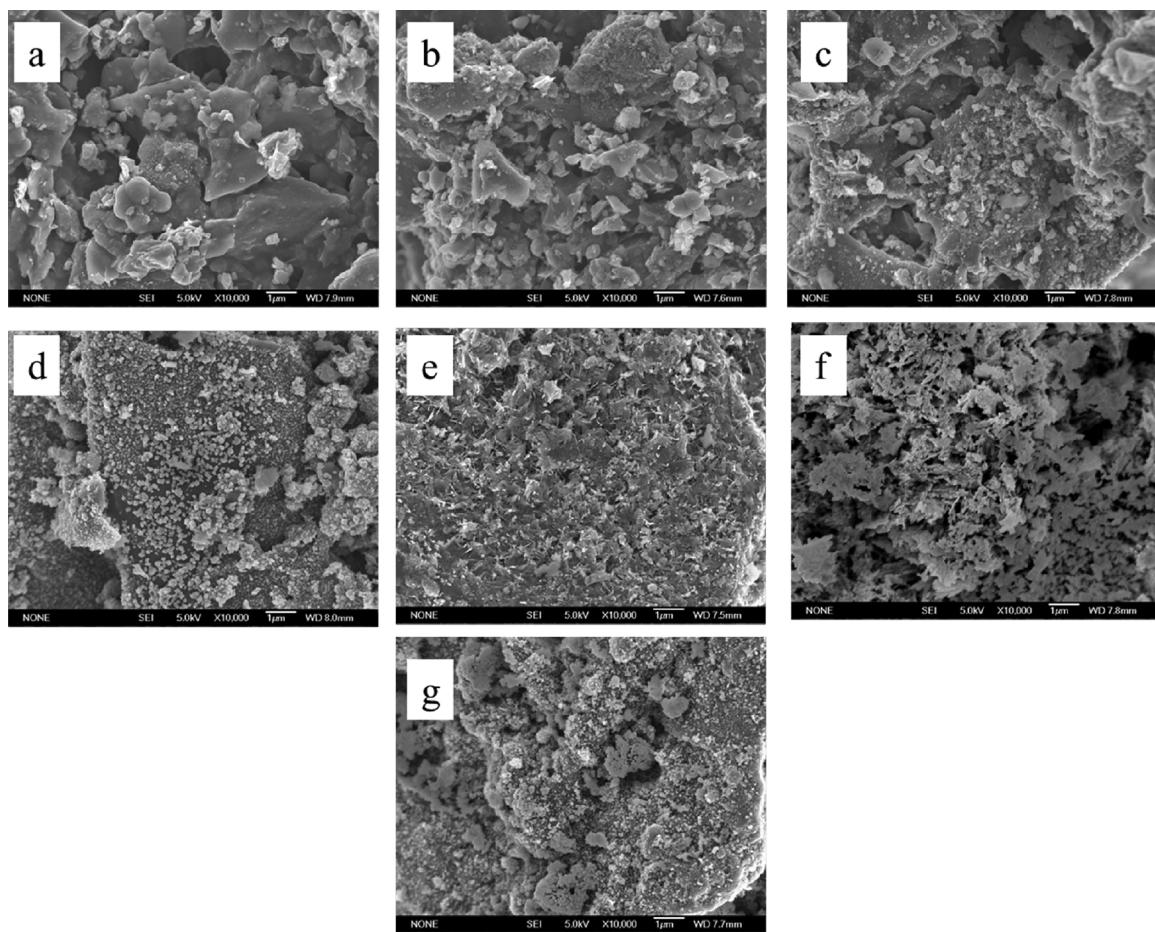


Fig. 2. SEM images ( $\times 10,000$ ) of (a) virgin AC, (b)  $\text{Mn}_{0.75}\text{Fe}_{0.65}/\text{AC}$ , (c)  $\text{Mn}_{0.75}\text{Fe}_{1.97}/\text{AC}$ , (d)  $\text{Mn}_{0.75}\text{Fe}_{3.30}/\text{AC}$ , (e)  $\text{Mn}_{0.75}\text{Fe}_{4.65}/\text{AC}$ , (f)  $\text{Mn}_{0.75}\text{Fe}_{6.02}/\text{AC}$ , (g)  $\text{Mn}_{0.75}\text{Fe}_{7.42}/\text{AC}$ .

exhibited another two peaks centered at 275 and 500 °C, representing the initial reduction of amorphous  $\text{MnO}_2$  to  $\text{Mn}_3\text{O}_4$  and the further reduction of  $\text{Mn}_3\text{O}_4$  to  $\text{MnO}$ , as well as the reduction of surface oxygen, while the broad higher peak at 675 °C might be partly contributed to the reduction of lattice oxygen [47]. For  $\text{Fe}_{6.02}/\text{AC}$  sample, the peak at 380 °C can be attributed to the reduction of  $\text{Fe}_2\text{O}_3$  to  $\text{Fe}_3\text{O}_4$ , whereas the higher temperature reduction peak at 550 °C owed to the transition of  $\text{Fe}_3\text{O}_4$  to  $\text{FeO}$  and the reduction of surface oxygen [3]. Compared with  $\text{Mn}_{6.02}/\text{AC}$  and  $\text{Fe}_{6.02}/\text{AC}$ , four peaks of  $\text{Mn}_{0.75}\text{Fe}_{6.02}/\text{AC}$  appeared at approximately 265, 470, 560 and 675 °C, corresponding to a superposition of the individual peaks of them. The peaks ascribed to the reduction of  $\text{MnO}_2 \rightarrow \text{Mn}_2\text{O}_3 \rightarrow \text{MnO}$  shifted to lower temperature at 265 and 470 °C, respectively. And the peak at 560 °C might be the result of the reduction of  $\text{Fe}^{3+}$  and the surface oxygen. It could be seen that the reduction peaks shifted towards lower temperature, manifesting the strong interaction of  $\text{MnO}_x\text{-FeO}_x$ , as well as the interaction between active components and support. In addition, the strong interaction between  $\text{MnO}_x$  and  $\text{FeO}_x$  can improve the mobility of active oxygen species over the surface of catalysts, which is conducive for the destruction of VOCs and the further oxidation of byproducts. The lower reduction temperature of sample is, the higher reducibility the sample owns, causing the formation of more oxygen vacancies [53]. Therefore,  $\text{Mn}_{0.75}\text{Fe}_{6.02}/\text{AC}$  was endowed with the capacity of producing more oxygen vacancies with the aid of  $\text{FeO}_x$  than  $\text{Mn}_{6.02}/\text{AC}$ . And then gas-phase  $\text{O}_2$  was easier to fill the oxygen vacancies, thus generating more active surface species [54].

### 3.2.2. XPS

The chemical states of the surface elements (C, O, Mn, Fe) over samples were elucidated by XPS measurement. Fig. 5A illustrates the

C1s XPS spectra. The characteristic peaks occurring in graphitic carbon (284.6–285.1 eV), carbon presenting in alcohol or ether groups (286.3–287.0 eV), carbonyl groups (287.5–288.1 eV), carboxyl or ester groups (289.3–290.0 eV) and shake-up satellite peaks due to  $\pi\text{-}\pi^*$  transitions in aromatic rings (291.2–292.1 eV) could be found [55]. With metal oxides doping, all peaks shifted to higher binding energy, and the ratio of C–C and O–C=O increased from 57.19% and 1.81% for virgin AC to 74.55% and 3.97%, respectively, while C–O– and C=O showed a decrease of 18.72% and 2.34% (Table 4). The acidity of  $\text{Mn}(\text{NO}_3)_2$  and  $\text{Fe}(\text{NO}_3)_3$  solution during the impregnation may give rise to the generation of acidic O–C=O groups, as reported in the literature [56]. Besides, the heating of AC under  $\text{N}_2$  atmosphere during calcination led to a drop in the concentration of surface oxygen functional groups caused by desorption of surface chemisorbed oxygen, thus an increase of C–C.

Additionally, the XPS spectra for O1s of samples could be fitted in three peaks at 529.2–530.2 eV, 531.3–532.3 eV and 532.7–533.5 eV, corresponding to lattice oxygen ( $\text{O}_L$ ), surface adsorbed oxygen, hydroxyl group and oxygen vacancies ( $\text{O}_{\text{OH}}$ ), and surface oxygen in adsorbed water species ( $\text{O}_W$ ) [57,58], respectively. Compared with virgin AC, the peaks of  $\text{O}_L$  and  $\text{O}_{\text{OH}}$  of modified AC shifted to higher binding energy, which might be contributed to the changes in the chemical environment of oxygen atoms after the loading of metal oxides [59]. In other words, a part of C atoms bonding with O were replaced by Mn and/or Fe atoms. Nevertheless, the co-modified AC owned lower binding energy of  $\text{O}_L$  than  $\text{Mn}_{6.02}/\text{AC}$  and  $\text{Fe}_{6.02}/\text{AC}$ , which acted as an oxygen storage role in metal oxide system [59]. Besides, the ratio of  $\text{O}_{\text{OH}}/\text{O}_T$  ( $\text{O}_T = \text{O}_L + \text{O}_{\text{OH}} + \text{O}_W$ ) over  $\text{Mn}_{0.75}\text{Fe}_{6.02}/\text{AC}$  (57.51%) was higher than that of  $\text{Mn}_{6.02}/\text{AC}$  (37.13%) and  $\text{Fe}_{6.02}/\text{AC}$  (43.08%) (Table 4), indicating the interaction between  $\text{MnO}_x$  and  $\text{FeO}_x$  was in

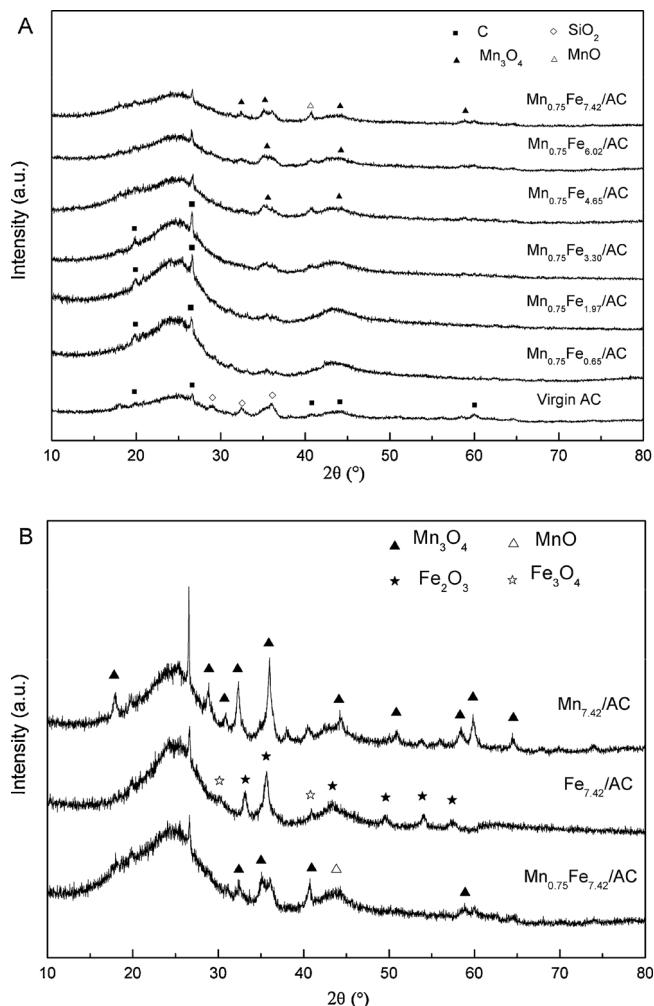


Fig. 3. XRD patterns of (A) Mn<sub>0.75</sub>Fe<sub>y</sub>/AC, (B) Mn<sub>7.42</sub>/AC, Fe<sub>7.42</sub>/AC and Mn<sub>0.75</sub>Fe<sub>7.42</sub>/AC.

favor of the generation of O<sub>OH</sub> on the surface of samples, which was consistent with H<sub>2</sub>-TPR (Fig. 4). On one hand, the surface adsorbed oxygen could facilitate the adsorption of gas-phase oxygen from simulated flue gas on the surface with high activity and mobility [60]. On the other hand, the generation of –OH, via the covalence bond between

H and surface adsorbed oxygen, was beneficial for the adsorption and oxidation of HCHO [53].

As seen from the Mn2p core spectra in Fig. 5C, for Mn<sub>6.02</sub>/AC, the peaks at 641.36 eV, 642.92 eV, 645.88 eV are ascribe to Mn<sup>2+</sup>, Mn<sup>3+</sup> and Mn<sup>4+</sup>, respectively [8]. After the doping of FeO<sub>x</sub>, all peaks shifted to lower binding energy, and the ratios of Mn<sup>4+</sup>/Mn<sup>3+</sup> and (Mn<sup>4+</sup> + Mn<sup>3+</sup>)/Mn<sup>2+</sup> increased (Table 4), suggesting that the incorporation of FeO<sub>x</sub> could facilitate the generation of Mn atoms with high valance on sample surface. As reported [61], Mn<sup>4+</sup> can directly oxidize the adsorbed reactant, and the Mn<sup>3+</sup> also plays an important role in oxidation with the presence of O<sub>2</sub>. Therefore, the higher values of (Mn<sup>4+</sup> + Mn<sup>3+</sup>)/Mn<sup>2+</sup> and Mn<sup>4+</sup>/Mn<sup>3+</sup> are favorable for HCHO oxidation.

As shown in Fig. 5D, two main asymmetric peaks located at 711.0 and 725.0 eV correspond to Fe2p<sub>3/2</sub> and Fe2p<sub>1/2</sub>, respectively. For Fe2p<sub>3/2</sub>, the binding energies at 710.2–710.4 eV and 711.3–712.1 eV are ascribed to the Fe<sup>3+</sup> cations in different forms, and another peak at 712.3–713.3 eV is attributed to Fe<sup>III</sup>-OH. Besides, the shake-up satellite peak at about 719.0 eV is the fingerprint of Fe<sup>3+</sup>. The results revealed that Fe<sup>3+</sup> was the only oxidation state existing on sample surface, while Fe<sup>2+</sup> existed inside the catalysts could not be detected by XPS characterization [8]. After the doping of MnO<sub>x</sub>, the ratio of Fe<sup>3+</sup>/Fe<sup>III</sup>-OH decreased from 1.53 to 1.17 (Table 4), suggesting the addition of MnO<sub>x</sub> over Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC induced the formation of Fe<sup>3+</sup> bonded with hydroxyl groups, which was in well agreement with the increase of O<sub>OH</sub>.

### 3.3. Catalytic performance

The removal performances of HCHO over Mn<sub>x</sub>Fe<sub>3.30</sub>/AC in simulated flue gas were investigated at the temperature of 150–330 °C. As shown in Fig. 6, virgin AC showed the lowest HCHO removal efficiency (E<sub>HCHO</sub>) of 60–70%. Obviously, the addition of MnO<sub>x</sub> or/and FeO<sub>x</sub> over AC significantly enhanced the HCHO removal activity of catalysts, and a promotional impact was obtained over Mn<sub>x</sub>Fe<sub>3.30</sub>/AC. E<sub>HCHO</sub> initially increased with the increasing temperature from 150 °C to 300 °C, especially a marked increase during 210–270 °C, and then leveled off finally. A higher temperature endowed the reactants with more kinetic energy, thus promoted the oxidation of HCHO. Among the catalysts, Mn<sub>0.75</sub>Fe<sub>3.30</sub>/AC exhibited the best performance in HCHO removal, reaching excellent removal efficiency above 85% with a broad reaction temperature window of 210–330 °C and the maximum of 96.75% at 300 °C. Furthermore, compared with single metal oxides doping, Mn<sub>x</sub>Fe<sub>3.30</sub>/AC was more effective for the removal of HCHO.

As a crucial factor, the content of metal oxides affects not only dispersion and aggregation of Mn<sub>x</sub>Fe<sub>y</sub>/AC, but also its crystallinity and

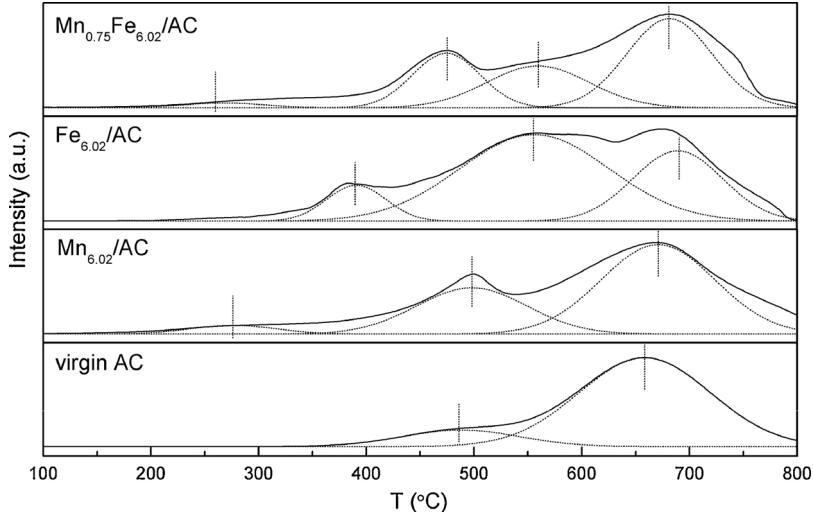
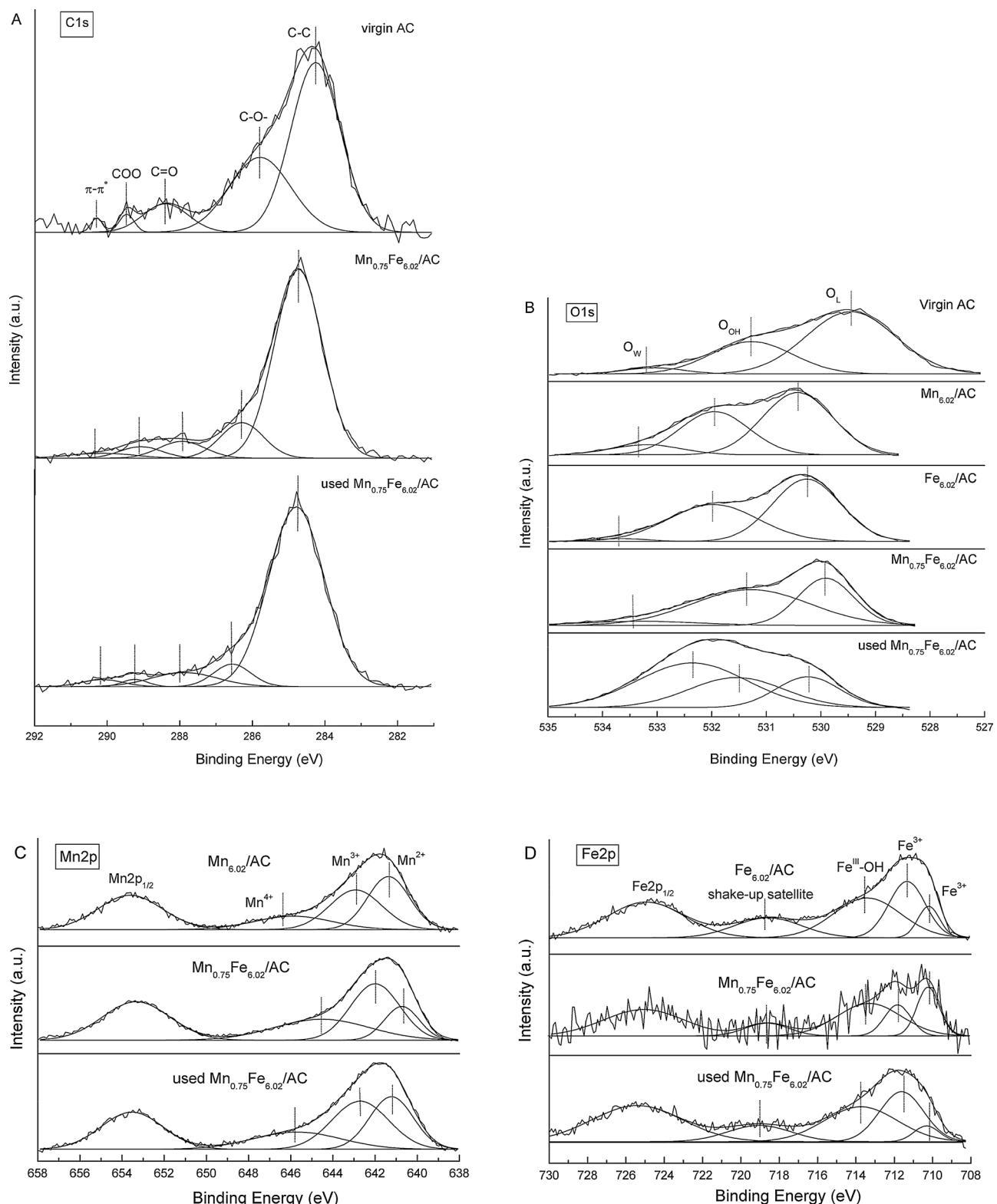


Fig. 4. H<sub>2</sub>-TPR profiles of virgin AC, Mn<sub>6.02</sub>/AC, Fe<sub>6.02</sub>/AC and Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC.



**Fig. 5.** XPS spectra for (A) C1s, (B) O1s, (C) Mn2p and (D) Fe2p of the samples before and after the reaction. (Reaction condition: 120 ppm HCHO, 6% O<sub>2</sub>, total flow rate 500 mL/min, reaction temperature 300 °C).

redox property. Therefore, the effect of loading value (mass percent of metal Mn-Fe on Mn<sub>x</sub>Fe<sub>y</sub>/AC) on E<sub>HCHO</sub> was evaluated (Fig. 7). It could be seen that the loading value significantly improved the HCHO removal efficiency. In particular, E<sub>HCHO</sub> initially increased remarkably, and then reached a maximum before a rapid decrease with further increase of loading value. In particular, for Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC, the catalyst,

with the best catalytic activity, exhibited the high E<sub>HCHO</sub> of 98.30%, which was 8.51% higher than Mn<sub>0.75</sub>Fe<sub>7.42</sub>/AC.

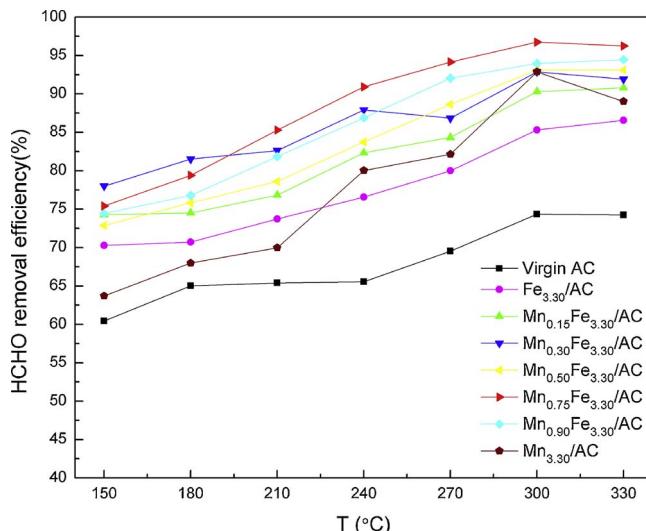
Obviously, Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC exhibited the best catalytic performance with the highest specific surface area and well-dispersed Mn-Fe oxides, which could provide more active sites for both adsorption and oxidation of HCHO. Besides, from the results of H<sub>2</sub>-TPR, the catalytic

**Table 4**  
The relative XPS intensity of as-prepared samples.

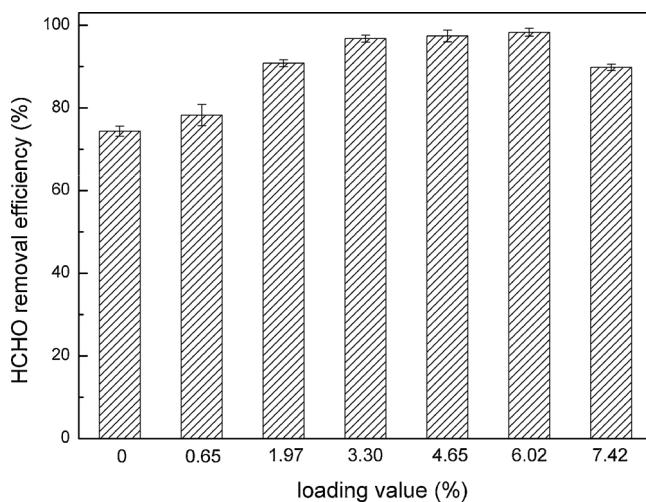
Relative intensity	AC	Mn <sub>6.02</sub> /AC	Fe <sub>6.02</sub> /AC	Mn <sub>0.75</sub> Fe <sub>6.02</sub> /AC	Used Mn <sub>0.75</sub> Fe <sub>6.02</sub> /AC
C-C(%)	57.19	–	–	74.55	81.86
C-O(%)	31.22	–	–	12.50	6.84
C = O(%)	8.69	–	–	6.35	7.50
COO(%)	1.81	–	–	3.97	1.48
π-π <sup>∗</sup> (%)	1.09	–	–	2.32	2.32
O <sub>L</sub> (%)	64.78	54.05	55.00	35.98	20.09
O <sub>OH</sub> (%)	30.98	37.13	43.08	57.51	29.55
O <sub>W</sub> (%)	4.24	8.83	1.92	6.51	50.36
<sup>a</sup> Mn <sup>2+</sup> (%)	–	29.39	–	13.72	25.91
<sup>a</sup> Mn <sup>3+</sup> (%)	–	27.47	–	32.55	30.84
<sup>a</sup> Mn <sup>4+</sup> (%)	–	13.77	–	23.37	17.26
Mn <sup>4+</sup> /Mn <sup>3+</sup>	–	0.50	–	0.72	0.56
Mn <sup>4+</sup> + Mn <sup>3+</sup> /Mn <sup>2+</sup>	–	1.40	–	4.08	1.86
<sup>b</sup> Fe <sup>3+</sup> (%)	–	–	42.97	36.93	38.41
<sup>b</sup> Fe <sup>III</sup> -OH(%)	–	–	28.07	31.39	26.93
Fe <sup>3+</sup> /Fe <sup>III</sup> -OH	–	–	1.53	1.17	1.43

<sup>a</sup> the percent of different valence of Mn in Mn2p<sub>3/2</sub>.

<sup>b</sup> the percent of different state of Fe in Fe2p<sub>3/2</sub> and shake-up satellite.



**Fig. 6.** Removal of HCHO on Mn<sub>x</sub>Fe<sub>3.30</sub>/AC samples in simulated flue gas. (Reaction condition: 120 ppm HCHO, 6% O<sub>2</sub>, 20 g sample, total flow rate 500 mL/min).



**Fig. 7.** Effect of loading value (y) on conversion of HCHO over Mn<sub>0.75</sub>Fe<sub>y</sub>/AC. (Reaction condition: 120 ppm HCHO, 6% O<sub>2</sub>, 20 g sample, total flow rate 500 mL/min, reaction temperature 300 °C).

activity of HCHO is closely related with the reducibility and oxygen vacancies. In combination with XPS characterization, for Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC, the high reducibility is contributed to the improvement in the mobility and amount of surface active oxygen (O<sub>L</sub>, O<sub>OH</sub>, etc.), and then facilitates a facile redox process with lower energy demand, thus leading to a promotional catalytic oxidation [62].

On the other hand, with the co-modification of Mn-Fe oxides, the increase of the amount of Mn<sup>4+</sup>, Mn<sup>3+</sup> and Fe<sup>III</sup>-OH enhanced the adsorption and oxidation of reactants. Besides, the synergistic effect in Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC induced the generation of surface active oxygen with high mobility and activity, thus leading to highest reducibility relative to the release of oxygen to react with HCHO and intermediates. Consequently, Mn<sub>0.75</sub>Fe<sub>y</sub> exhibited excellent HCHO catalytic efficiency above 90% at 300 °C as the loading value was more than 1.97%.

### 3.4. Effect of gas components

#### 3.4.1. Effect of O<sub>2</sub>

As a crucial gas component, effect of O<sub>2</sub> on HCHO removal and CO<sub>2</sub> conversion over Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC was investigated. As shown in Fig. 8, at the temperature window of 60–150 °C, the HCHO removal efficiency decreased in all three atmospheres, and the CO<sub>2</sub> conversion increased with the increasing temperature. The results above indicated that the physisorption played predominant role at low temperature and was gradually substituted by chemisorption as the temperature increased. And for higher temperature, catalytic oxidation, instead of adsorption, became the main factor influencing HCHO removal as more kinetic energy was provided by increasing temperature, which could be deduced from the higher efficiency of HCHO removal and CO<sub>2</sub> conversion with the presence of O<sub>2</sub>. At the absence of O<sub>2</sub>, Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC still possessed relatively high activity at high temperature by the consumption of abundant surface active oxygen. This may be because that the co-modification of MnO<sub>x</sub> and FeO<sub>x</sub> could improve the redox properties and the mobility of the active oxygen species over the surface of catalysts. After the addition of 6% O<sub>2</sub>, both E<sub>HCHO</sub> and S<sub>C</sub> increased sharply, while there was no evident increase of E<sub>HCHO</sub> and S<sub>C</sub> with 12% O<sub>2</sub>/N<sub>2</sub>. The results above indicated that gas-phase O<sub>2</sub> could regenerate the lattice oxygen and replenish the chemisorbed oxygen, and 6% O<sub>2</sub> was sufficient for the HCHO oxidation.

#### 3.4.2. Effect of SO<sub>2</sub>

The slight inhibitory effect of SO<sub>2</sub> on HCHO removal was observed (Fig. 9). E<sub>HCHO</sub> were 57.35% and 52.87% with respect to 400 and 800 ppm SO<sub>2</sub> in the absence of O<sub>2</sub>, respectively, which declined slightly compared with that of pure N<sub>2</sub>. As a polar molecule, SO<sub>2</sub> (dipole

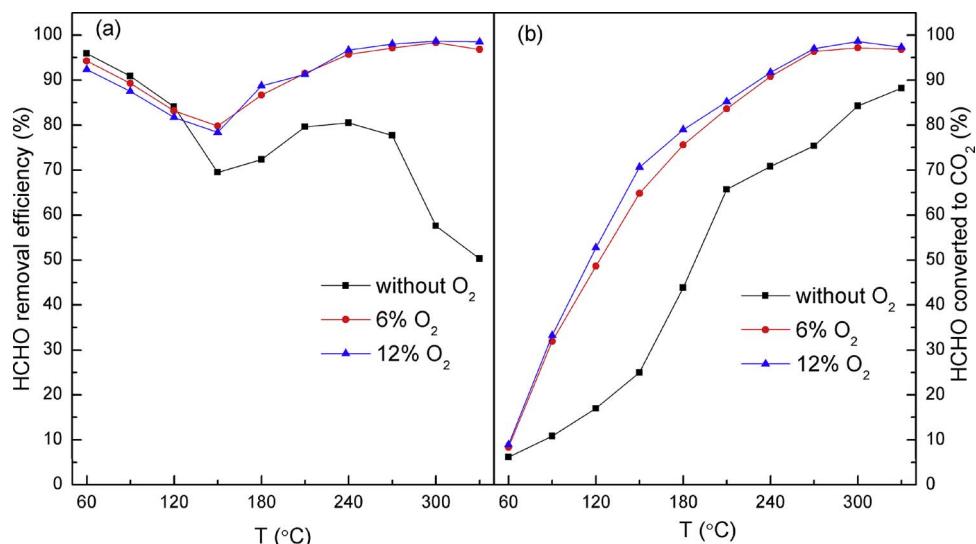


Fig. 8. HCHO removal (a) and HCHO converted to CO<sub>2</sub> (b) as a function of reaction temperature in simulated flue gas with different O<sub>2</sub> concentration for Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC. (Reaction condition: 120 ppm HCHO, 0–12% O<sub>2</sub>, 20 g sample, total flow gas rate 500 mL/min).

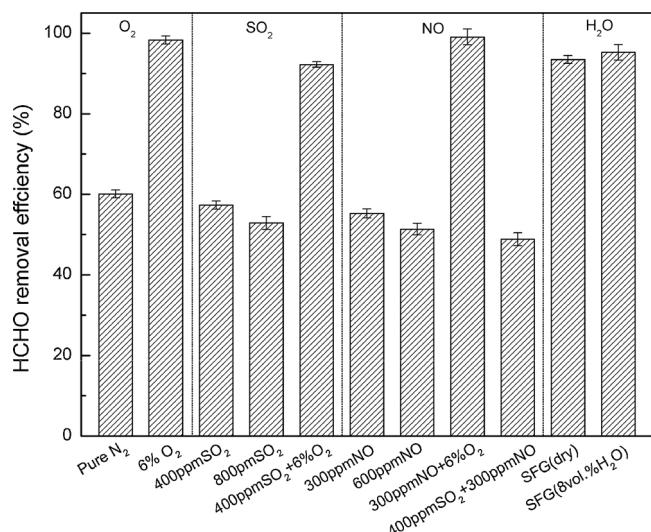


Fig. 9. Effect of flue gas components on HCHO removal over Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC. (Reaction condition: 120 ppm HCHO, 0–6% O<sub>2</sub>, 0–800 ppm SO<sub>2</sub>, 0–600 ppm NO, 0–8 vol.% H<sub>2</sub>O, 20 g sample, total flow rate 500 mL/min, reaction temperature 300 °C, and SFG = 120 ppm HCHO + 300 ppm NO + 400 ppm SO<sub>2</sub> + 6% O<sub>2</sub> + N<sub>2</sub>).

moment = 1.60D) leaned to be adsorbed by some polar or basic adsorption sites, whereas compared with the strong polarity of HCHO (dipole moment = 2.33D), there was less competition with HCHO for active sites. At the same time, SO<sub>2</sub> or SO<sub>3</sub> oxidized by surface oxygen also could react with metal oxides over AC and formed sulfite and sulfate, which not only reduced the effective active components for HCHO oxidation, but caused the blockage of micropores for the adsorption of reactant [60]. With the aid of 6% O<sub>2</sub>, the inhibition of SO<sub>2</sub> was weakened as a result of the replenishment of the surface oxygen consumed by SO<sub>2</sub>.

#### 3.4.3. Effect of NO

As shown in Fig. 9, HCHO removal efficiencies were 55.26% and 51.35% in the presence of 300 and 600 ppm NO balanced in N<sub>2</sub>, respectively. As we all known, with lower diameter, NO molecules could spread more quickly than HCHO to occupy active sites for the following adsorption and oxidation [64]. Moreover, NO could be oxidized by the limited surface oxygen easily to form NO<sub>2</sub>, even NO<sub>2</sub><sup>–</sup> and NO<sub>3</sub><sup>–</sup> [65]. Although NO<sub>2</sub> possesses higher oxidizing power than O<sub>2</sub> for the

oxidation of VOCs, as reported in literature [65], NO still inhibited HCHO removal over Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC with the absence of O<sub>2</sub> because of the overwhelming limit from NO taking up active sites and consuming active oxygen. Therefore, introducing 6% O<sub>2</sub> into the flue gas with 300 ppm NO balanced in N<sub>2</sub>, E<sub>HCHO</sub> was enhanced significantly to 99.05% higher than that of the absence of NO. The replenishment of active oxygen due to the presence of gas-phase O<sub>2</sub> produced more NO<sub>2</sub> as a more powerful oxidant, thus replaced or assisted O<sub>2</sub> for the reoxidation of the reduced metal oxides sites faster through the recycle of NO<sub>2</sub> ↔ NO + O<sup>\*</sup> [66]. In a word, O<sub>2</sub> played an important role in the effect of NO on HCHO removal.

When SO<sub>2</sub> and NO coexisted in pure N<sub>2</sub>, only 48.85% HCHO was removed due to the stronger competition for active sites among the reactants. With the presence of 6% O<sub>2</sub>, a high HCHO removal efficiency of 93.45% was reached over Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC, demonstrating that Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC has great potential in practical application. It has been reported [67] that the reaction (NO)<sub>ads</sub> + (O<sub>2</sub>)<sub>ads</sub> + (SO<sub>2</sub>)<sub>ads</sub> → [(NO<sub>2</sub>) (SO<sub>3</sub>)]<sub>ads</sub> promoted the adsorption of NO and SO<sub>2</sub>, and then strengthened the competition for active sites between HCHO, NO and SO<sub>2</sub>.

#### 3.4.4. Effect of H<sub>2</sub>O

To investigate the effect of water vapor on HCHO removal, 8 vol.% water vapor was added into the simulated flue gas (SFG) at optimal reaction temperature. As seen from Fig. 9, the addition of 8 vol.% water vapor increased the E<sub>HCHO</sub> from 93.45% to 95.26%. The slight increase might be contributed to the activation of surface active oxygen (O<sub>2</sub><sup>–</sup>, O<sup>–</sup>, etc) by H<sub>2</sub>O according to the reaction O<sub>2</sub><sup>–</sup>, O<sup>–</sup> + H<sub>2</sub>O → O<sup>\*</sup> + 2–OH, through which way the consumed hydroxyl groups could be regenerated. Chen et al. [14] also reported the positive effect of H<sub>2</sub>O both on the formation and further oxidation of the formate intermediates, leading to an enhanced HCHO removal activity, while Raupp et al. [68] found the water vapor would block the unsaturated coordination sites via the associative and dissociative (as –OH) adsorption on reduced cations, thus leading to the dramatic loss of the catalytic activity. The results above suggest the catalyst possesses excellent H<sub>2</sub>O durability, which is competitive in practical application on catalytic removal of HCHO, and the specific mechanism how SO<sub>2</sub>, NO and H<sub>2</sub>O affect the HCHO oxidation is expected to explore in our further study.

To measure the stability and selectivity of Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC in the removal of HCHO from SFG, the oxidation of HCHO at 300 °C was conducted for 18 h, and the results are presented in Fig. 10. E<sub>HCHO</sub> increased sharply from 80.35% to 95.37% in initial 8 h and then kept stable at above 95% except for a negligible fluctuation. It is worth

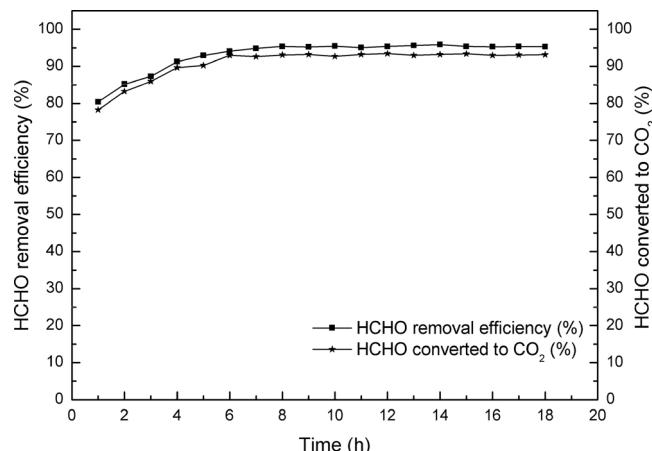


Fig. 10. Stability test of HCHO conversion over Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC. (Reaction condition: 120 ppm HCHO, 300 ppm NO, 400 ppm SO<sub>2</sub>, 6% O<sub>2</sub>, 8 vol.% H<sub>2</sub>O and balance N<sub>2</sub>, 20 g sample, total flow rate 500 mL/min, reaction temperature 300 °C).

noting that the variation of CO<sub>2</sub> conversion (S<sub>C</sub>) was similar to that observed in HCHO removal, and slightly lower than HCHO removal efficiency. This suggested that a small amount of byproducts generated in HCHO oxidation, such as formate and carbonate intermediates, which agreed with the following FT-IR results and previous study [63]. It was manifested that Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC owned an excellent stability and selectivity for the removal of HCHO from simulated flue gas, as well as good resistance to SO<sub>2</sub>, NO and H<sub>2</sub>O, which was significant in practical applications.

### 3.5. Simultaneous removal of Hg<sup>0</sup> and HCHO

The simultaneous removal of Hg<sup>0</sup> and HCHO over Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC were investigated at the temperature of 150–330 °C, and the results were shown in Fig. 11. It is obvious that Hg<sup>0</sup> removal efficiency decreased with the increasing temperature from 150 to 210 °C and then increased sharply to reach the highest removal efficiency of 77.51% at 270 °C in the presence of HCHO. As shown in Fig. 11B, the removal of mercury over samples occurred as the synergy of adsorption and oxidation, where adsorption acted as the removal mechanism at lower temperature, and then was substituted by the oxidation with the increasing reaction temperature, which was in accordance with Zhao et. al [69]. Besides, oxidation was in the lead of HCHO removal, and increased as the temperature increased. Therefore, 270 °C was chosen as the optimal temperature for the simultaneous removal of Hg<sup>0</sup> and HCHO. Furthermore, a connection between Hg<sup>0</sup> and HCHO removal was also studied. It was obvious that the addition of HCHO significantly inhibited the removal of mercury due to the competition of active sites. However, after introducing Hg<sup>0</sup>, a slight decrease lower than 6% for EHCHO was observed, which might be the mercury concentration (90 µg/m<sup>3</sup>) was extremely too small compared with HCHO concentration (120 ppm).

### 3.6. Mechanism discussion

The surface of Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC before and after exposure to HCHO was investigated by XPS. As shown in Fig. 5A, after treated with HCHO, C=O showed a slight increase compared with the fresh sample, which might be ascribe to the adsorption of HCHO on samples and/or the oxidation of HCHO to formate species. And the decrease of C–O and COO, as well as the increase of C–C, indicated the participation of surface adsorbed oxygen during the oxidation of HCHO. The results above elucidated the oxygen-containing functional groups and π–π\* interactions not only played an important role in the coating of active components on the surface of AC, but also participated in the oxidation

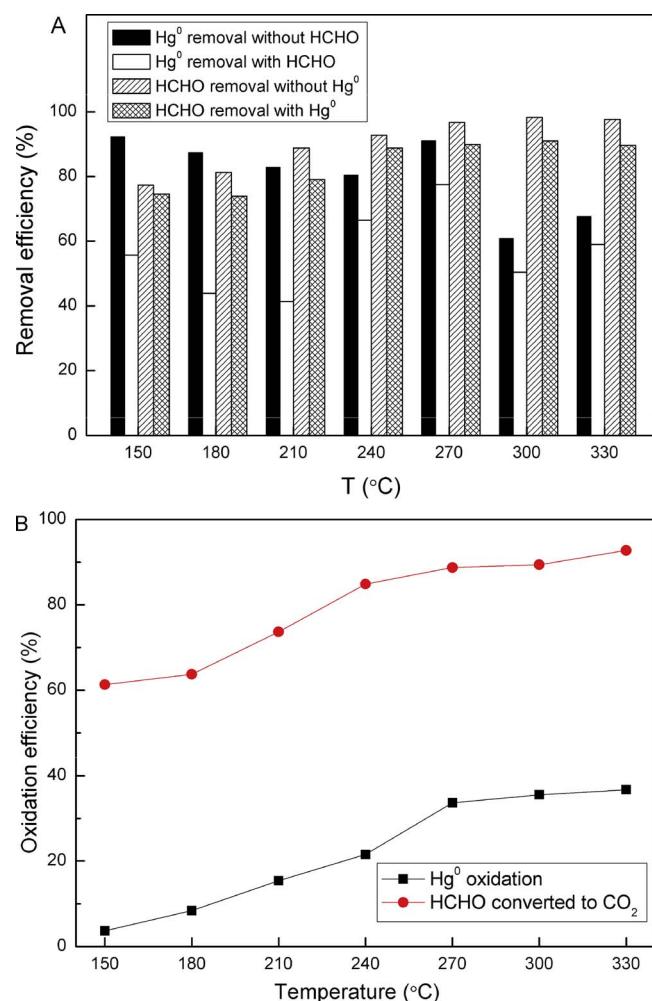


Fig. 11. Simultaneous (A) removal and (B) oxidation of Hg<sup>0</sup> and HCHO over Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC. (Reaction condition: 120 ppm HCHO, 90 µg/m<sup>3</sup> Hg<sup>0</sup>, 6% O<sub>2</sub>, 20 g sample, total flow rate 500 mL/min).

of HCHO. For O1s, the ratio of O<sub>L</sub> and O<sub>OH</sub> over used Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC exhibited a sharp decrease, especially for O<sub>OH</sub>, indicating that both O<sub>L</sub> and O<sub>OH</sub> participated in HCHO removal. Xu et al. [70] also reported that abundant hydroxyls could offer more adsorption sites for HCHO, and facilitate the oxidation of formate species into CO<sub>2</sub> and H<sub>2</sub>O. Besides, an increase of 43.85% for the ratio of O<sub>W</sub> revealed that an amount of H<sub>2</sub>O was generated during HCHO decomposition and oxidation.

The oxidation states of the metal elements over fresh and used Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC were also studied. Comparing with fresh Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC, Mn<sup>4+</sup> and Mn<sup>3+</sup> decreased sharply, especially for Mn<sup>4+</sup>, while more Mn<sup>2+</sup> was generated after treated with the reaction (Table 4), demonstrating that the reaction of Mn<sup>4+</sup> → Mn<sup>3+</sup> → Mn<sup>2+</sup> occurred, and then facilitated the oxidation of HCHO. Besides, the ratio of Fe<sup>3+</sup>/Fe<sup>III</sup>-OH for Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC increased from 1.17 to 1.43 (Table 4), indicating that the –OH bonded with Fe<sup>3+</sup> participated in the oxidation of HCHO, further confirming the promotion of –OH in the process of HCHO removal. It has also been reported that the promotion of Fe<sup>III</sup>-OH was attributed to the hydrogen-bonding interaction between the H atoms of Fe<sup>III</sup>-OH and the O atoms of HCHO molecules [71].

To illustrate the oxidation mechanism of HCHO over Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC, the FT-IR spectra of Mn<sub>0.75</sub>Fe<sub>6.02</sub>/AC with exposure to HCHO with different atmosphere and different temperatures were investigated (Fig. 12). Upon exposure to 120 ppm HCHO/N<sub>2</sub> at room temperature for 2 h (curve b), bands at 3747, 2300–2400, 1575, 1134, 600–700 cm<sup>-1</sup> were observed. According to literatures, the band at 1134 cm<sup>-1</sup> was corresponded to molecularly adsorbed formaldehyde [72,73], which

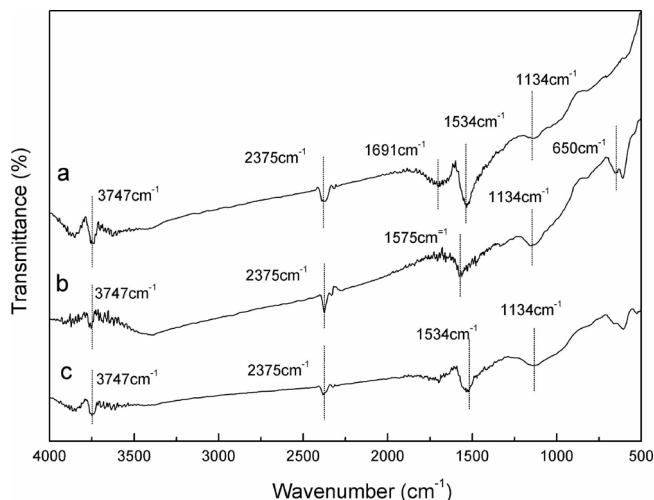


Fig. 12. FT-IR spectra of  $\text{Mn}_{0.75}\text{Fe}_{6.02}/\text{AC}$  under different conditions: a. 120 ppm HCHO + 6%  $\text{O}_2 + \text{N}_2$  at  $300\text{ }^\circ\text{C}$ ; b. 120 ppm HCHO +  $\text{N}_2$  at room temperature; c. 120 ppm HCHO + 6%  $\text{O}_2 + \text{N}_2$  at room temperature.

suggested that HCHO adsorbed on  $\text{Mn}_{0.75}\text{Fe}_{6.02}/\text{AC}$  catalyst surface firstly. The bands related to the formation of formate species adsorbed on the catalyst surface at 3747, 1575 and 600–700  $\text{cm}^{-1}$  were assigned to the isolated –OH of formic acid, the asymmetric stretching vibration of  $\text{COO}$  ( $\nu_{\text{as}}(\text{COO})$ ) of formate and strong C–H deformation, respectively [14,74]. Moreover, the peak of adsorbed  $\text{CO}_2$  (bands at 2300–2400  $\text{cm}^{-1}$ ) might be partly generated during HCHO oxidation process, which was further proved by  $\text{CO}_2$  detected in outlet flue gas. It indicated that the  $\text{Mn}_{0.75}\text{Fe}_{6.02}/\text{AC}$  could act as an efficient catalyst with sufficient oxygen functional groups for the oxidation of HCHO even in the absence of  $\text{O}_2$  [14]. Upon introducing  $\text{O}_2$  into the feed stream (curve c), similar bands were observed. However, the band at 1575  $\text{cm}^{-1}$  ( $\nu_{\text{as}}(\text{COO})$ ) disappeared and the relative intensity of bands at 600–700  $\text{cm}^{-1}$  was weakened. And the band at 1534  $\text{cm}^{-1}$  was assigned to the asymmetric vibration of carbonate species [74], indicative of that the gas-phase  $\text{O}_2$  facilitated the further oxidation of formate species through the supplement of surface active oxygen. For curve a, the intensity of band at 1134  $\text{cm}^{-1}$  weakened and band at

1534  $\text{cm}^{-1}$  strengthened. Besides, the band at 1691  $\text{cm}^{-1}$  was also ascribed to carbonate species. With the increase of temperature, part of adsorbed HCHO could be oxidized to carbonate species rapidly, and then facilitated the oxidation of HCHO into  $\text{CO}_2$ , which was consistent with the experiment results.

A possible mechanism of HCHO removal on the  $\text{Mn}_x\text{Fe}_y/\text{AC}$  is proposed based on the obtained results and previous studies, as shown in Fig. 13. With the co-modification of Mn–Fe oxides,  $\text{Mn}_x\text{Fe}_y/\text{AC}$  owned more active surface oxygen, more Mn atoms with high valence and  $\text{Fe}^{3+}$  bonded with –OH, and the good reducibility due to the synergy of  $\text{MnO}_x$  and  $\text{FeO}_x$ . Firstly, HCHO molecule was adsorbed on the surface of catalyst via the hydroxyl groups and some other active sites, and abundant active surface oxygen ( $\text{O}^\bullet$ ) released via the redox cycles of  $\text{Fe}^{3+}/\text{Fe}^{2+}$  and  $\text{Mn}^{4+}/\text{Mn}^{3+}/\text{Mn}^{2+}$ , as well as chemisorbed oxygen of surface functional groups over activated coke. Secondly, the nucleophilic  $\text{O}^\bullet$  attracted C–H of adsorbed formaldehyde molecule to produce the formate species, which was proved by the peaks ascribed to C–H deformation and formate species in FT-IR, as well as the C1s XPS spectra. Meanwhile, the oxygen vacancy was formed and then filled via the decomposition of gas phase  $\text{O}_2$  molecule rapidly. Subsequently, the active hydroxyl radicals further oxidized formate intermediate into adsorbed  $\text{CO}_2$  and  $\text{H}_2\text{O}$ . During the process, part of formate species were converted into carbonate species after losing another –H and then were further oxidized into  $\text{CO}_2$  and  $\text{H}_2\text{O}$ , which was also detected in FT-IR characterization. Moreover, the lost –H could react with the unbonded –OH adsorbed on the sample surface to generate  $\text{H}_2\text{O}$ . Finally, the generated  $(\text{CO}_2)_{\text{ads}}$  and  $(\text{H}_2\text{O})_{\text{ads}}$  desorbed from the sample surface.

#### 4. Conclusion

The  $\text{Mn}_x\text{Fe}_y/\text{AC}$  samples were prepared by impregnation method for the HCHO removal. The superior HCHO removal efficiency above 90% was obtained over  $\text{Mn}_{0.75}\text{Fe}_{3.30}/\text{AC}$  at the temperature window of 240–330  $^\circ\text{C}$ , and the maximum of that reached 98.30% over  $\text{Mn}_{0.75}\text{Fe}_{6.02}/\text{AC}$  at  $300\text{ }^\circ\text{C}$ . Besides,  $\text{Mn}_{0.75}\text{Fe}_{6.02}/\text{AC}$  also exhibited excellent stability and selectivity during 18 h test, and had potential in the simultaneous removal of HCHO and  $\text{Hg}^0$ .  $\text{SO}_2$  and  $\text{NO}$  showed an inhibitory effect on HCHO removal without  $\text{O}_2$  due to the competitive adsorption. However, gas-phase  $\text{O}_2$  regenerated the lattice oxygen and

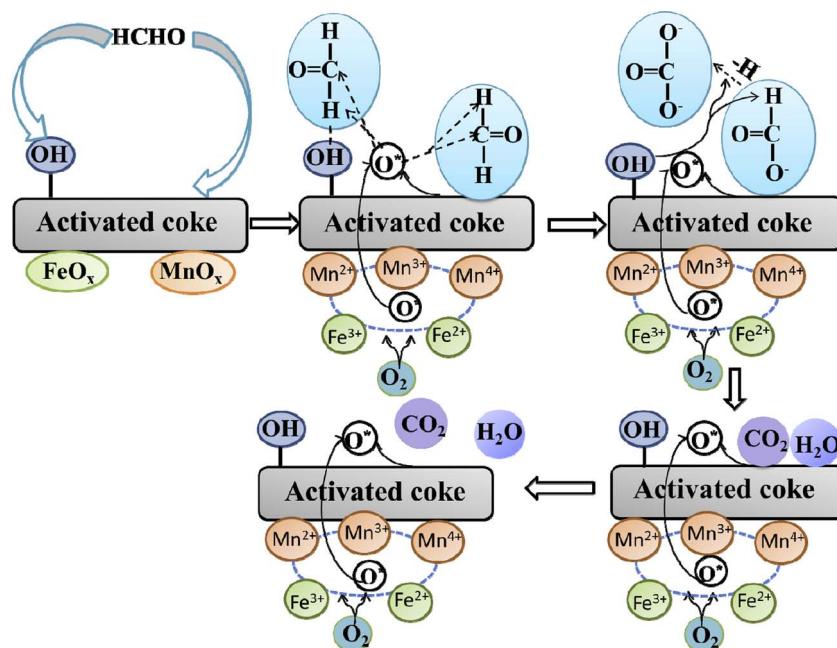


Fig. 13. The proposed mechanism of HCHO removal process over  $\text{Mn}_x\text{Fe}_y/\text{AC}$ .

supplemented the chemisorbed oxygen, facilitating the removal of HCHO. Therefore, with the addition of O<sub>2</sub>, the negative effect of SO<sub>2</sub> was weakened, and NO promoted HCHO removal due to the generation of abundant NO<sub>2</sub>, which possessed higher oxidizing power than O<sub>2</sub>. H<sub>2</sub>O further enhanced the HCHO oxidation via the supplement of the consumed –OH. In addition, characterization results demonstrated that Mn<sub>0.75</sub>Fe<sub>0.02</sub>/AC possessed large surface area, high dispersion of active components and higher reducibility, which facilitated the removal of HCHO. XPS results indicated that oxygen-containing groups, such as C–O- and COO-, abundant active surface oxygen and –OH, played an important role in both adsorption and oxidation of HCHO, as well as the redox cycle of Fe<sup>3+</sup>/Fe<sup>2+</sup> and Mn<sup>4+</sup>/Mn<sup>3+</sup>/Mn<sup>2+</sup>. Moreover, based on FT-IR results, HCHO was adsorbed on the samples via –OH and other active sites, and then oxidized into formate and carbonate intermediates, finally further oxidized into CO<sub>2</sub> and H<sub>2</sub>O. Further studies to investigate the mechanism how NO, SO<sub>2</sub> and H<sub>2</sub>O affect HCHO removal and the simultaneous removal of HCHO and Hg<sup>0</sup> are warranted.

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## References

- C. Deng, Q. Huang, X. Zhu, Q. Hu, W. Su, J. Qian, L. Dong, B. Li, M. Fan, C. Liang, *Appl. Surf. Sci.* 389 (2016) 1033–1049.
- S. Lu, K. Li, F. Huang, C. Chen, B. Sun, *Appl. Surf. Sci.* 400 (2017) 277–282.
- B.B. Chen, X.B. Zhu, M. Crocker, Y. Wang, C. Shi, *Appl. Catal. B: Environ.* 154–155 (2014) 73–81.
- L. Ma, D. Wang, J. Li, B. Bai, L. Fu, Y. Li, *Appl. Catal. B: Environ.* 148–149 (2014) 36–43.
- J. Yu, X. Li, Z. Xu, W. Xiao, *Environ. Sci. Technol.* 47 (2013) 9928–9933.
- M. Chidambaram, S. Selvakumar, T. Tamil Selvi, A.P. Singh, *J. Mol. Catal. A* 245 (2006) 69–77.
- G.K. Zhang, X. Qin, *Mater. Res. Bull.* 48 (2013) 3743–3749.
- P. Liu, H. He, G. Wei, X. Liang, F. Qi, F. Tan, W. Tan, J. Zhu, R. Zhu, *Appl. Catal. B: Environ.* 182 (2016) 476–484.
- W.J. Liang, J. Li, J.X. Li, T. Zhu, Y.Q. Jin, *J. Hazard. Mater.* 175 (2010) 1090–1095.
- Z. Xu, N. Qin, J. Wang, H. Tong, *Bioresour. Technol.* 101 (2010) 6930–6934.
- J.I. Gutierrez-Ortiz, B. de Rivas, R. Lopez-Fonseca, S. Martin, J.R. Gonzalez-Velasco, *Chemosphere* 68 (2007) 1004–1012.
- L. Qi, B. Cheng, J. Yu, W. Ho, *J. Hazard. Mater.* 301 (2016) 522–530.
- D. Li, G. Yang, P. Li, J. Wang, P. Zhang, *Catal. Today* (2016).
- B.B. Chen, C. Shi, M. Crocker, Y. Wang, A.M. Zhu, *Appl. Catal. B: Environ.* 132–133 (2013) 245–255.
- Z. Tang, W. Zhang, Y. Li, Z. Huang, H. Guo, F. Wu, J. Li, *Appl. Surf. Sci.* 364 (2016) 75–80.
- S. Colussi, M. Boaro, L. de Rogatis, A. Pappacena, C. de Leitenburg, J. Llorca, A. Trovarelli, *Catal. Today* 253 (2015) 163–171.
- B.B. Chen, X.B. Zhu, M. Crocker, Y. Wang, C. Shi, *Catal. Commun.* 42 (2013) 93–97.
- B. Liu, Y. Liu, C. Li, W. Hu, P. Jing, Q. Wang, J. Zhang, *Appl. Catal. B: Environ.* 127 (2012) 47–58.
- X. Tang, J. Chen, X. Huang, Y. Xu, W. Shen, *Appl. Catal. B: Environ.* 81 (2008) 115–121.
- Y. Yu, T. Wang, Y. Fu, W. Su, J. Hu, *Int. J. Hydrogen Energy* 39 (2014) 17617–17621.
- Y. Ma, G. Zhang, *Chem. Eng. J.* 288 (2016) 70–78.
- H. Chen, Z. Rui, X. Wang, H. Ji, *Catal. Today* 258 (2015) 56–63.
- G.K. Zhang, Q. Xiong, W. Xu, S. Guo, *Appl. Clay. Sci.* 102 (2014) 231–237.
- S. Shin, J. Song, *J. Hazard. Mater.* 194 (2011) 385–392.
- Q. Wen, C. Li, Z. Cai, W. Zhang, H. Gao, L. Chen, G. Zeng, X. Shu, Y. Zhao, *Bioresour. Technol.* 102 (2011) 942–947.
- W.D.P. Rengga, A. Chafidz, M. Sudibandriyo, M. Nasikin, A.E. Abasaeed, *J. Environ. Chem. Eng.* 5 (2017) 1657–1665.
- J. Li, N. Kobayashi, Y. Hu, *Chem. Eng. Process.* 47 (2008) 118–127.
- K. Jastrz̄ab, *Fuel Process. Technol.* 101 (2012) 16–22.
- F. Sun, J. Gao, X. Liu, X. Tang, S. Wu, *Appl. Surf. Sci.* 357 (2015) 1895–1901.
- Y. Itaya, K. Kawahara, C.W. Lee, J. Kobayashi, N. Kobayashi, S. Hatano, S. Mori, *Fuel* 88 (2009) 1665–1672.
- S. Ogriseck, G.P.G. Vanegas, *Chem. Eng. J.* 160 (2010) 641–650.
- S.S. Tao, C.T. Li, X.P. Fan, G.M. Zeng, P. Lu, X. Zhang, Q.B. Wen, W.W. Zhao, D.Q. Luo, C.Z. Fan, *Chem. Eng. J.* 210 (2012) 547–556.
- H. Wu, C.T. Li, L.K. Zhao, J. Zhang, G. Zeng, Y. Xie, X. Zhang, Y. Wang, *Energy Fuel* 29 (2015) 6747–6757.
- Y. Wang, C.T. Li, L.K. Zhao, Y. Xie, X. Zhang, G. Zeng, H. Wu, J. Zhang, *Environ. Sci. Pollut. R.* 23 (2016) 5099–5110.
- K. Tong, A. Lin, G. Ji, D. Wang, X. Wang, *J. Hazard. Mater.* 308 (2016) 113–119.
- M.H. Zhang, Q.L. Zhao, Z.F. Ye, *J. Environ. Sci.* 23 (2011) 1962–1969.
- B. Zhao, X. Liu, Z. Zhou, H. Shao, M. Xu, *Chem. Eng. J.* 284 (2016) 1233–1241.
- Y. Xie, C. Li, L. Zhao, J. Zhang, G. Zeng, X. Zhang, W. Zhang, S. Tao, *Appl. Surf. Sci.* 333 (2015) 59–67.
- X. Peng, H. Yi, X. Tang, K. Li, *Res. Chem. Intermed.* 41 (2013) 213–222.
- S.A.C. Carabineiro, N. Bogdanchikova, P.B. Tavares, J.L. Figueiredo, *RSC Adv.* 2 (2012) 2957.
- X. Liang, P. Liu, H. He, G. Wei, T. Chen, W. Tan, F. Tan, J. Zhu, R. Zhu, *J. Hazard. Mater.* 306 (2016) 305–312.
- S.C. Kim, W.G. Shim, *Appl. Catal. B: Environ.* 98 (2010) 180–185.
- H. Tian, J. He, X. Zhang, L. Zhou, D. Wang, *Microporous Mesoporous Mater.* 138 (2011) 118–122.
- R. Lin, W.-P. Liu, Y.-J. Zhong, M.-F. Luo, *Appl. Catal. A: Gen.* 220 (2001) 165–171.
- F.F.A. Parmaliana, *Appl. Catal. A: Gen.* 226 (2002).
- J. He, Y. Li, D. An, Q. Zhang, Y. Wang, *J. Nat. Gas Chem.* 18 (2009) 288–294.
- L.K. Zhao, C.T. Li, S.H. Li, Y. Wang, J. Zhang, T. Wang, G. Zeng, *Appl. Catal. B: Environ.* 198 (2016) 420–430.
- C. He, B. Shen, G. Chi, F. Li, *Chem. Eng. J.* 300 (2016) 1–8.
- Y.C. Xie, Y.Q. Tang, *Adv. Catal.* 37 (1990) 1–43.
- M.H. Castaño, R. Molina, S. Moreno, *Appl. Catal. A: Gen.* 492 (2015) 48–59.
- J. Zhou, L. Zhao, Q. Huang, R. Zhou, X. Li, *Catal. Lett.* 127 (2008) 277–284.
- G. Zhang, Z. Li, H. Zheng, T. Fu, Y. Ju, Y. Wang, *Appl. Catal. B: Environ.* 179 (2015) 95–105.
- D. Li, G. Yang, P. Li, J. Wang, P. Zhang, *Cataly. Today* 277 (2016) 257–265.
- C. Ma, D. Wang, W. Xue, B. Dou, H. Wang, Z. Hao, *Environ. Sci. Technol.* 45 (2011) 3628–3634.
- S. Biniak, G. Szymbański, J. Siedlewski, A. Świtkowski, *Carbon* 35 (1997) 1799–1810.
- Y. Huang, J. Tang, L. Gai, Y. Gong, H. Guan, R. He, H. Lyu, *Chem. Eng. J.* 319 (2017) 229–239.
- J. Deng, S. He, S. Xie, H. Yang, Y. Liu, G. Guo, H. Dai, *Environ. Sci. Technol.* 49 (2015) 11089–11095.
- M.V. Gallegos, M.A. Peluso, E. Finocchio, H.J. Thomas, G. Busca, J.E. Sambeth, *Chem. Eng. J.* 313 (2017) 1099–1111.
- D. Jampaiah, S.J. Ippolito, Y.M. Sabri, B.M. Reddy, S.K. Bhargava, *Catal. Sci. Technol.* 5 (2015) 2913–2924.
- J.J. Sheng, C.T. Li, L.K. Zhao, X.Y. Du, L. Gao, G. Zeng, *Fuel* 197 (2017) 397–406.
- J.R. Li, J.S. Chen, Y.K. Yu, C. He, *J. Ind. Eng. Chem.* 25 (2015) 352–358.
- S.M. Li, Q.L. Hao, R.Z. Zhao, D.L. Liu, H.Z. Duan, B. J. Dou, *Chem. Eng. J.* 285 (2016) 536–543.
- G.Y. Popova, T.V. Andrushevich, E.V. Semionova, Y.A. Chesarov, L.S. Dovlitova, V.A. Rogov, V.N. Parmon, *J. Mol. Catal. A* 283 (2008) 146–152.
- H. Li, C.Y. Wu, Y. Li, L. Li, Y. Zhao, J. Zhang, *J. Hazard. Mater.* 243 (2012) 117–123.
- A. Aissat, D. Courcot, R. Cousin, S. Siffert, *Catal. Today* 176 (2011) 120–125.
- F. Bertinchamps, M. Treinen, N. Blangenois, E. Mariage, E. Gaigneaux, *J. Catal.* 230 (2005) 493–498.
- G.B. Raupp, J.A. Dumisec, *J. Phys. Chem.* 89 (1985) 5240–5246.
- Y. Guo, Y. Li, T. Zhu, M. Ye, *Energy Fuel* 27 (2013) 360–366.
- B. Zhao, H.H. Yi, X.L. Tang, Q. Li, D.D. Liu, F.Y. Gao, *Chem. Eng. J.* 286 (2016) 585–593.
- Z. Xu, J. Yu, M. Jaroniec, *Appl. Catal. B: Environ.* 163 (2015) 306–312.
- Z. Xu, J. Yu, W. Xiao, *Chem-Eur. J.* 19 (2013) 9592–9598.
- T. Kecskés, J. Raskó, J. Kiss, *Appl. Catal. A: Gen.* 273 (2004) 55–62.
- H. Li, G. Qi, X. Tana, X. Zhang, W. Huang, W. Li, Shen, *Appl. Catal. B: Environ.* 103 (2011) 54–61.
- G.B. Elisabetta Finocchio, *Catal. Today* 70 (2001) 213–225.